

**Recent Developments on
Radiation Hard Semiconductor Detectors for SuperLHC
- CERN-RD50 project -**

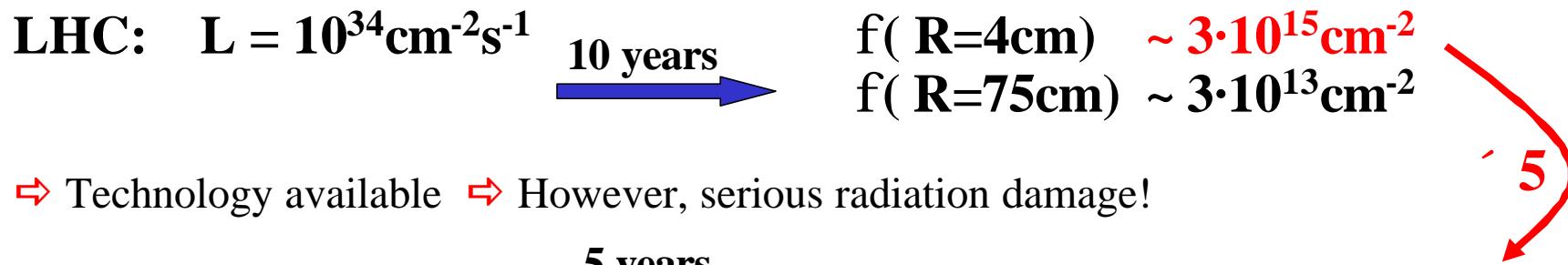
Michael Moll
on behalf of RD50
CERN - Geneva - Switzerland

OUTLINE

- Motivation to develop radiation harder detectors
- The RD50 collaboration
- Radiation Damage – a very brief review
- Approaches to obtain radiation hard sensors
 - Material engineering
 - Device engineering
- Summary

<http://www.cern.ch/rd50>

- **LHC upgrade** (“Super-LHC” ... later than 2010)



5

- **LHC experiments** (...starting 2007)

- ⇒ **Radiation hard technologies now adopted have not been completely characterized:** Oxygen-enriched Si in ATLAS/CMS pixels
- ⇒ **Replacement of components** e.g. for LHCb Velo at $r < 4\text{cm}$ a replacement of detectors is foreseen after 3 years operation

- **Linear collider experiments**

Deep understanding of radiation damage will be fruitful for linear collider experiments where high doses of e , γ will play a significant role.



RD50: Development of Radiation Hard Semiconductor Devices for High Luminosity Colliders

- Collaboration formed in November 2001
- Experiment approved as RD50 by CERN in June 2002
- Main objective:

Development of ultra-radiation hard semiconductor detectors for the luminosity upgrade of the LHC to $10^{35} \text{ cm}^{-2}\text{s}^{-1}$ (“Super-LHC”).

Challenges:

- **Radiation hardness up to 10^{16} cm^{-2} required**
- **Fast signal collection** (Going from 25ns to 10 ns bunch crossing ?)
- **Low mass** (reducing multiple scattering close to interaction point)
- **Cost effectiveness** (big surfaces have to be covered with detectors!)

- Presently 271 Members from 52 Institutes

Belarus (Minsk), Belgium (Louvain), Canada (Montreal), Czech Republic (Prague (2x)), Finland (Helsinki, Lappeenranta), Germany (Berlin, Dortmund, Erfurt, Hamburg, Karlsruhe), Israel (Tel Aviv), Italy (Bari, Bologna, Florence, Milano, Padova, Perugia, Pisa, Trento, Trieste, Turin), Lithuania (Vilnius), Norway (Oslo (2x)), Poland (Warsaw), Romania (Bucharest (2x)), Russia (Moscow (2x), St.Petersburg), Slovenia (Ljubljana), Spain (Barcelona, Valencia), Switzerland (CERN, PSI), Ukraine (Kiev), United Kingdom (Exeter, Glasgow, Lancaster, Liverpool, Sheffield, University of Surrey), USA (Fermilab, Purdue University, Rochester University, Rutgers University, SCIPP Santa Cruz, Syracuse University, BNL, University of New Mexico)

Scientific strategies:

I. Material engineering

II. Device engineering

III. Variation of detector operational conditions

CERN-RD39

“Cryogenic Tracking Detectors”

<http://cern.ch/rd39>

- Defect Engineering of Silicon

- Understanding radiation damage
 - Macroscopic effects and Microscopic defects
 - *Simulation of defect properties and defect kinetics*
 - *Irradiation with different type of particles at different energies*

- Oxygen enriched silicon

- DOFZ – Diffusion Oxygenated Float Zone Silicon
- Cz - Czochralski Silicon
- MCZ - Magnetic Czochralski
- EPI – Epitaxial silicon grown on CZ substrate

- *Oxygen dimer enriched silicon*

- *Hydrogen enriched silicon*
- *Pre-irradiated silicon*

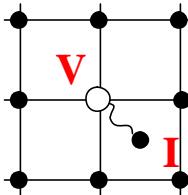
- New Materials

- Silicon Carbide (SiC)
- Gallium Nitride (GaN)
- (Diamond: CERN RD42 Collaboration)

- Device Engineering (New Detector Designs)

- Thin detectors
- 3D detectors
- Semi 3D detectors
- *Cost effective detectors*
- p-type silicon detectors
- *Simulation of highly irradiated detectors*

particle → Si_S → $E_K > 25 \text{ eV}$



Vacancy
+
Interstitial

point defects
(V-O, C-O, ..)

$E_K > 5 \text{ keV}$ point defects and clusters of defects

- **^{60}Co -gammas**

- Compton Electrons with max. $E_\gamma \approx 1 \text{ MeV}$ (no cluster production)

- **Electrons**

- $E_e > 255 \text{ keV}$ for displacement
- $E_e > 8 \text{ MeV}$ for cluster

- **Neutrons (elastic scattering)**

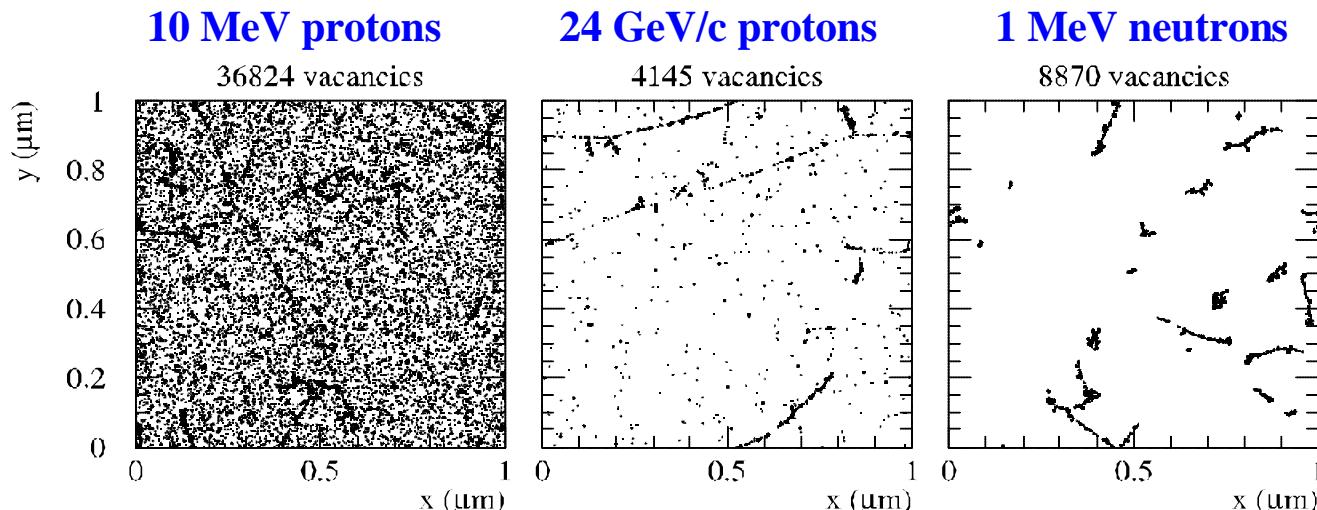
- $E_n > 185 \text{ eV}$ for displacement
- $E_n > 35 \text{ keV}$ for cluster

Only point defects \longleftrightarrow point defects & clusters \longleftrightarrow Mainly clusters

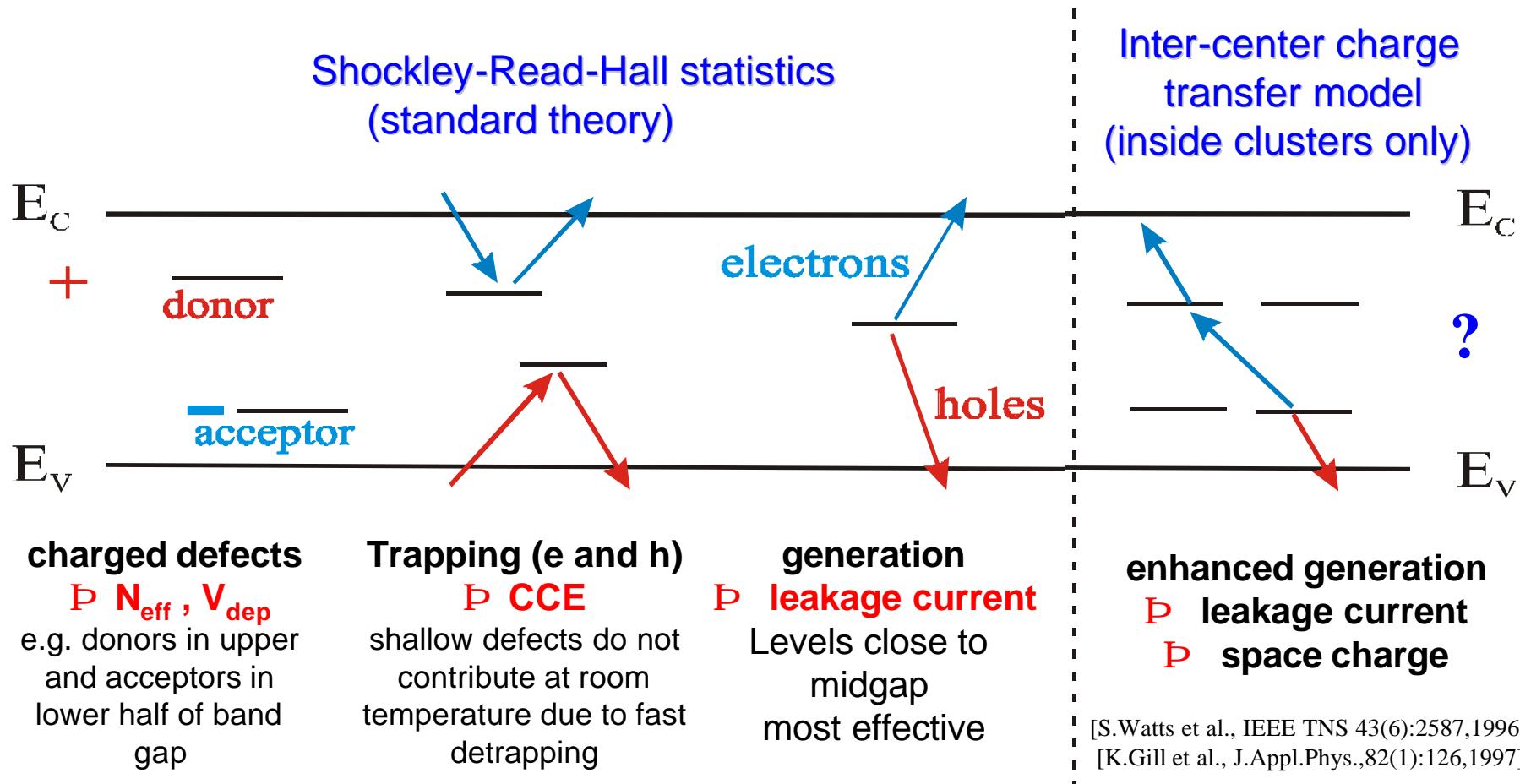
Simulation:

Initial distribution of vacancies in $(1\text{mm})^3$ after $10^{14} \text{ particles/cm}^2$

[Mika Huhtinen NIMA 491(2002) 194]



RD50 Impact of Defects on Detector properties

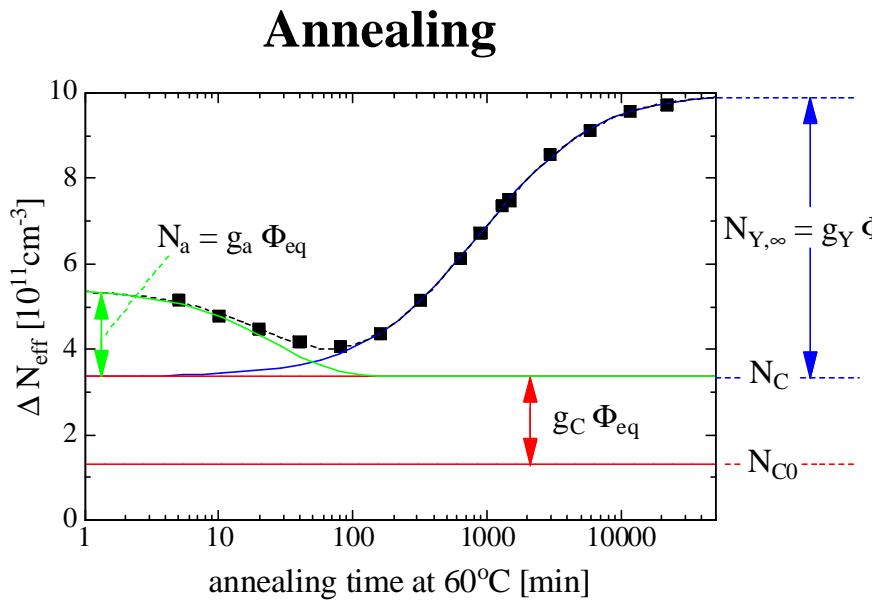
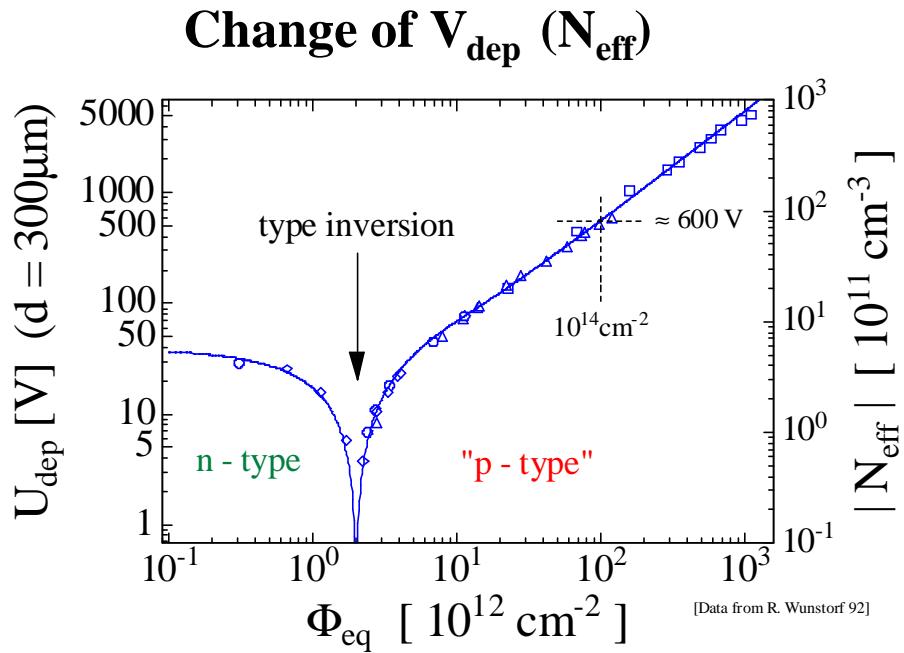


Impact on detector properties can be calculated if all defect parameters are known:

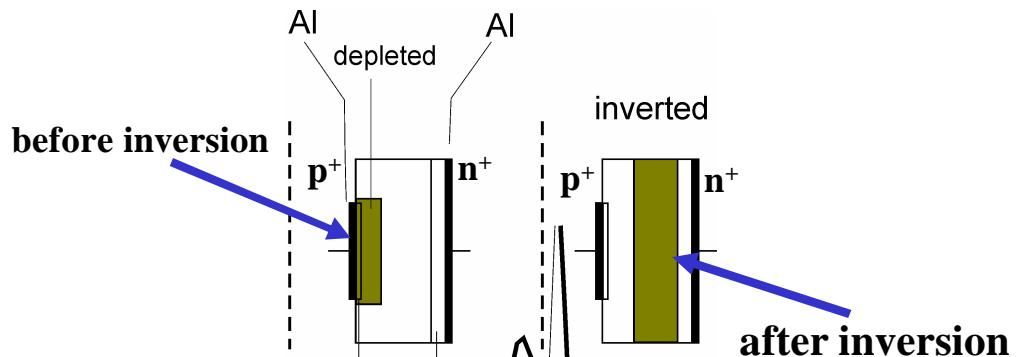
S_{n,p} : cross sections

DE : ionization energy

N_t : concentration



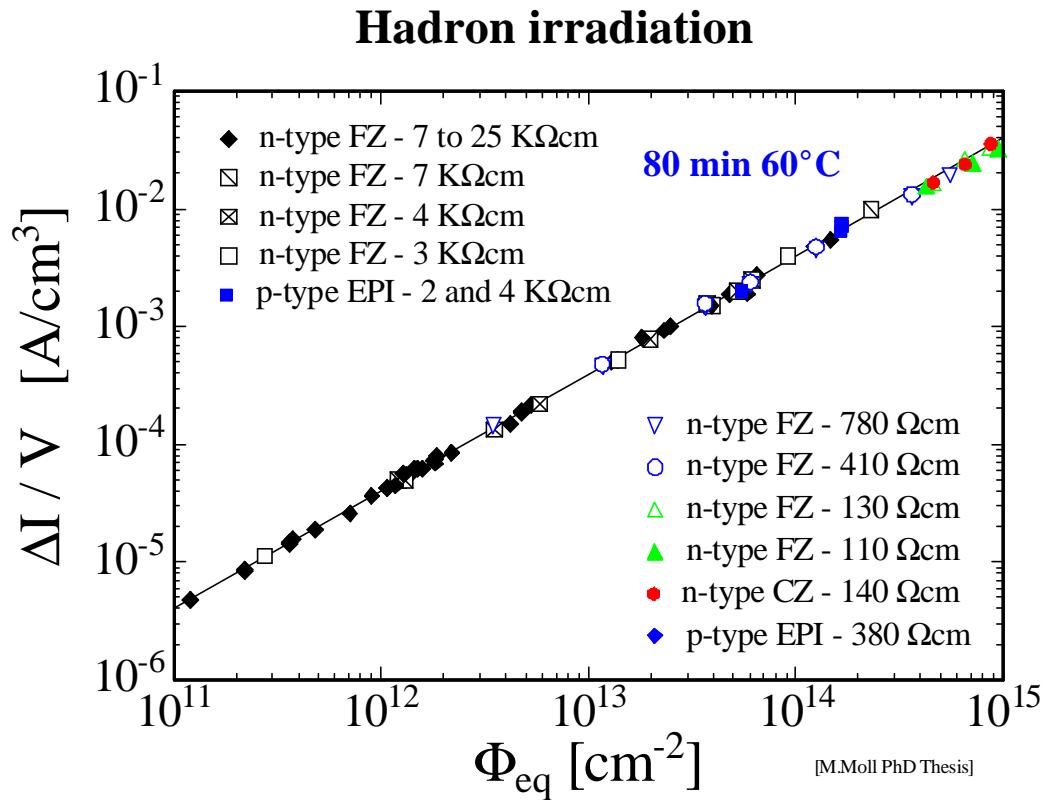
- Type inversion:
SCSI – Space Charge Sign Inversion



- Short term: “**Beneficial annealing**”
- Long term: “**Reverse annealing**”

time constant :

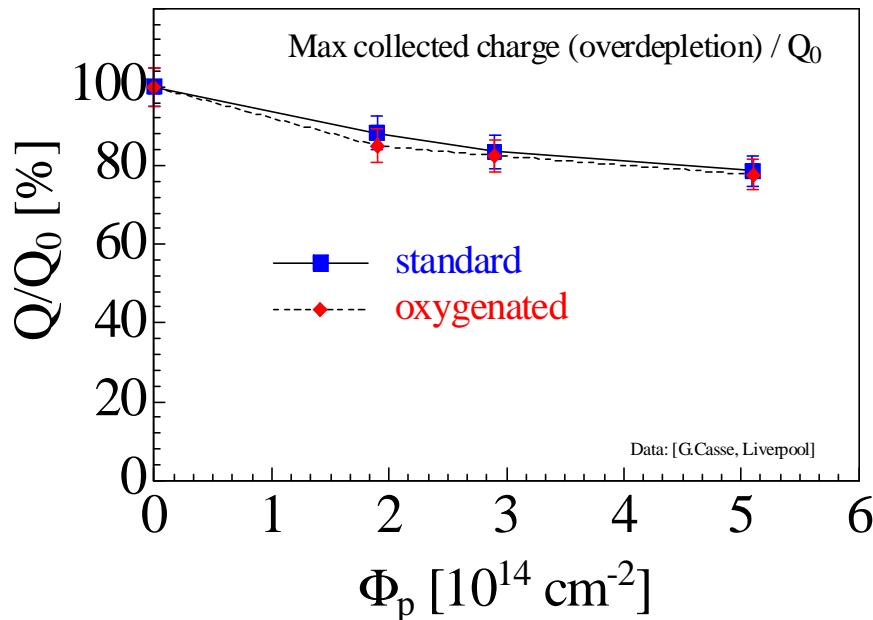
~ 500 years (-10°C)
~ 500 days (20°C)
~ 21 hours (60°C)



- Damage parameter α (slope) independent of fluence
- α independent of Φ_{eq} and impurities
 - ⇒ can be used for fluence calibration (NIEL-Hypothesis)
 - (take care about annealing effects!)

- **Mechanisms reducing collected charge:**
 - Trapping (electrons and holes)
 - Underdepletion (detector design and geometry)
 - Type inversion

ATLAS microstrip + RO electronics



- **Oxygenation has no influence on trapping.**
- **After $5 \cdot 10^{14} \text{ p/cm}^2$ (24GeV/c)**
 - 80% of charge collected (25ns)
 - overdepletion needed !

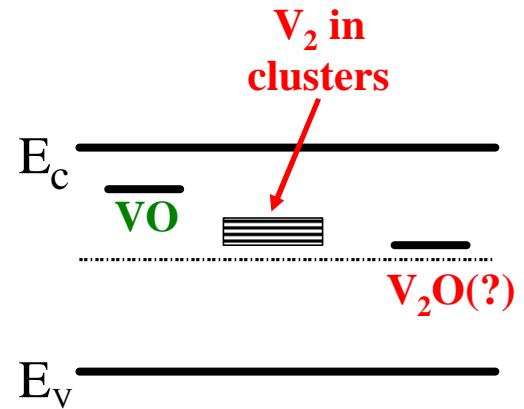
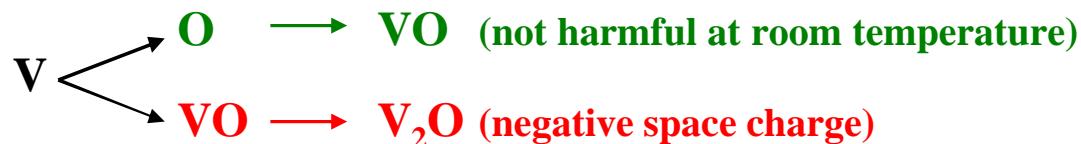
Data: Gianluigi Casse; 1st Workshop on Radiation Hard Semiconductor Devices for High Luminosity Colliders; CERN; 28-30 November 2002

- Influence the defect kinetics by incorporation of impurities or defects
- Best example: Oxygen

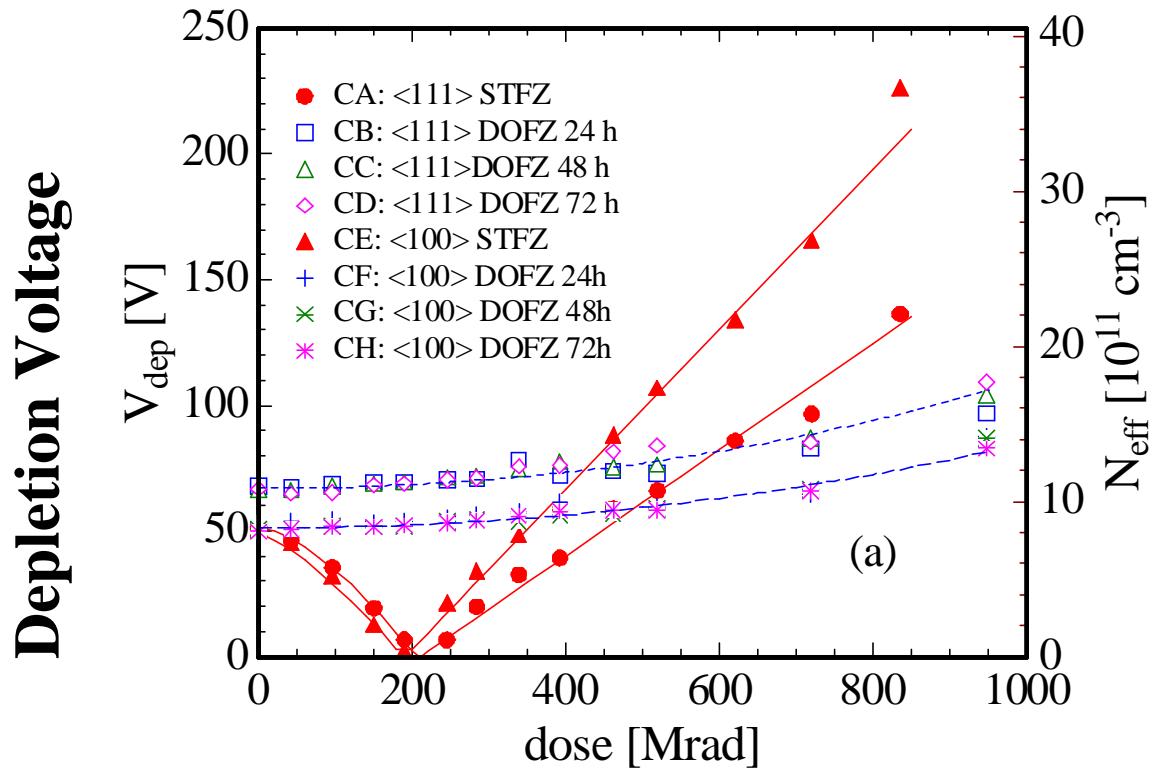
Initial idea: Incorporate Oxygen to getter radiation-induced vacancies
 P prevent formation of Di-vacancy (V_2) related deep acceptor levels

Observation: Higher oxygen content P less negative space charge
 (less charged acceptors)

- One possible mechanism: V_2O is a deep acceptor



Oxygen enriched silicon: Spectacular Improvement of g-irradiation tolerance



(a)

- No type inversion for oxygen enriched silicon!
- Slight increase of positive space charge
(due to Thermal Donor generation?)

[E.Fretwurst et al. 1st RD50 Workshop]

See also:

- Z.Li et al. [NIMA461(2001)126]

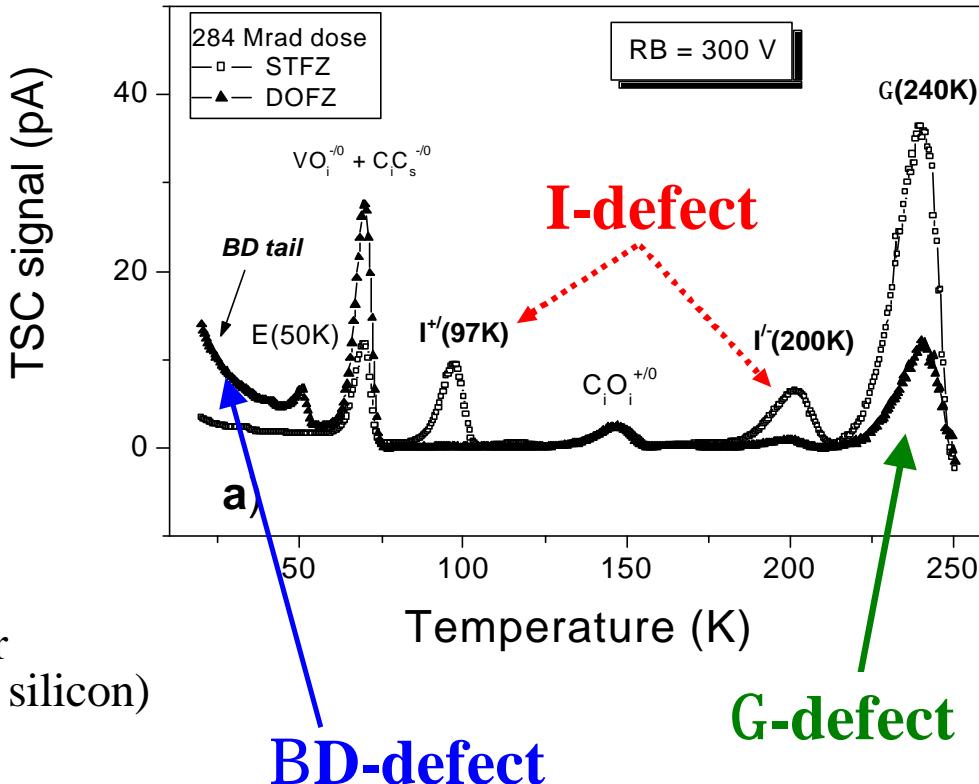
- Z.Li et al. [1st RD50 Workshop]

- **For the first time** macroscopic changes of the depletion voltage and leakage current can be explained by electrical properties of measured defects → **Major breakthrough!**

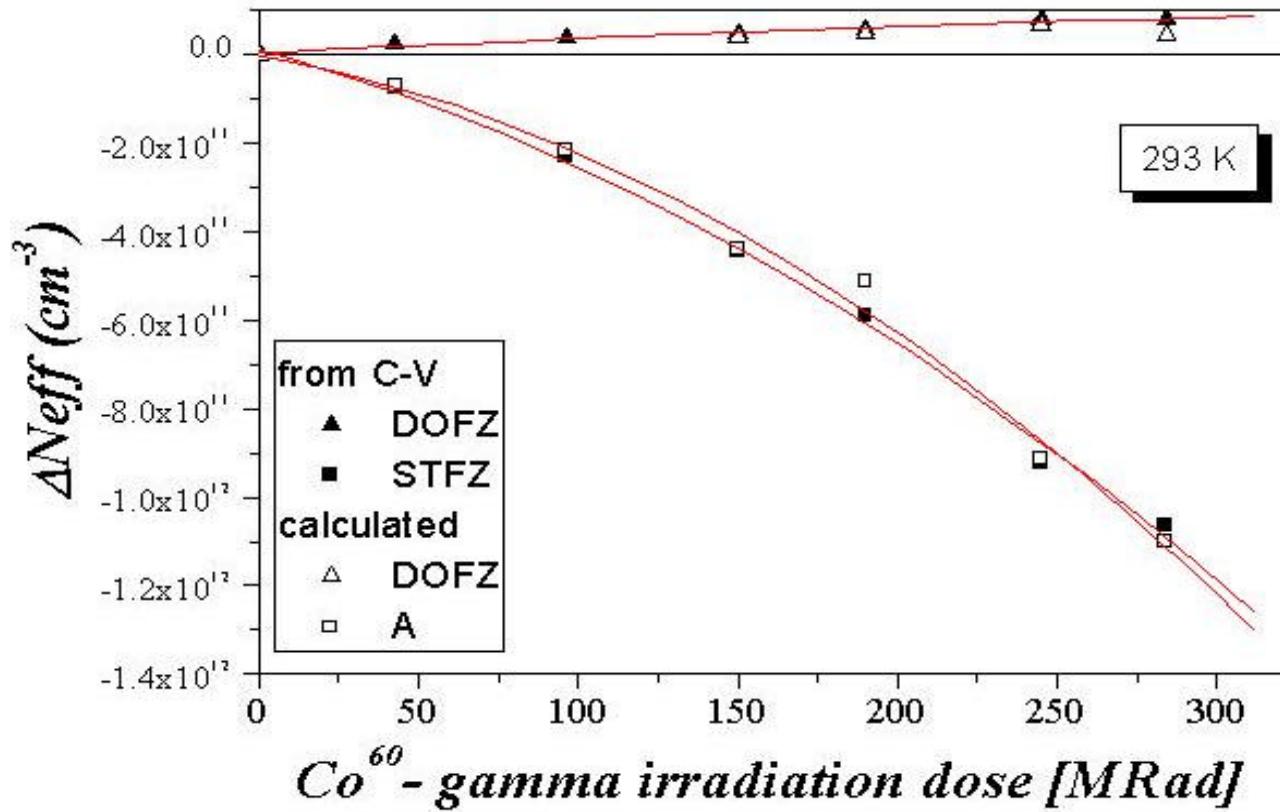
[Applied Physics Letters, 82, 2169, March 2003]

Levels responsible for macroscopic changes after g-irradiation:

- **I-defect:** acceptor level at $E_C - 0.54\text{eV}$
(coming up for approx. 85% of damage)
peculiarity: quadratic dose dependence
P good candidate for the V_2O defect
- **G-defect:** acceptor level at $E_V + 0.68\text{eV}$
(coming up for approx. 10% of damage)
- **BD-defect:** bistable shallow thermal donor
(important in oxygen enriched silicon)



- Comparison for effective doping concentration for two different materials
 - as predicted by the microscopic measurements (open symbols)
 - as deduced from CV characteristics (filled symbols)



[I.Pintilie et al.,
NIMA514, 18, 2003]

- Excellent agreement also for the increase of leakage current (not shown here)

- **DOFZ (Diffusion Oxygenated Float Zone Silicon)**

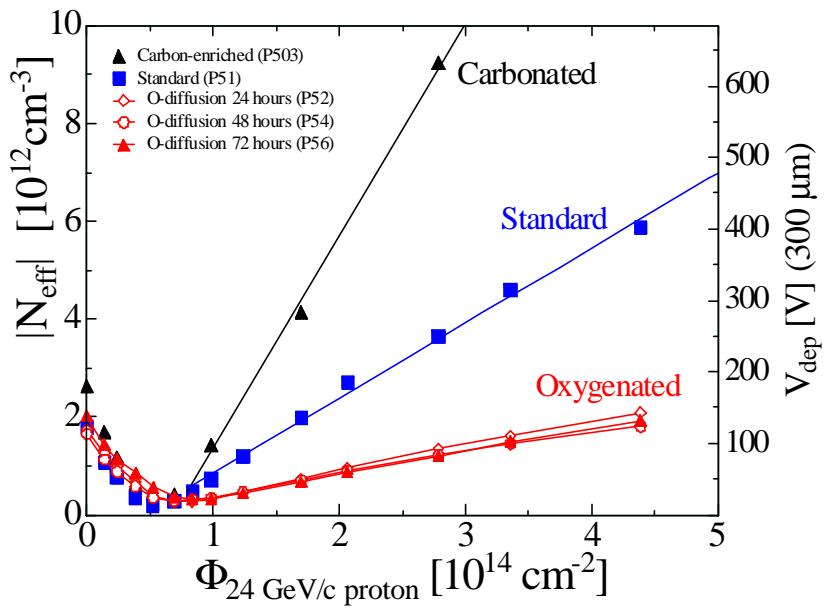
- 1982 First oxygen diffusion tests on FZ [Brotherton et al. J.Appl.Phys., Vol.53, No.8.,5720]
- 1995 First tests on detector grade silicon [Z.Li et al. IEEE TNS Vol.42, No.4, 219]
- **1999 Introduced to the HEP community by RD48 (ROSE)**

ROSE
RD48



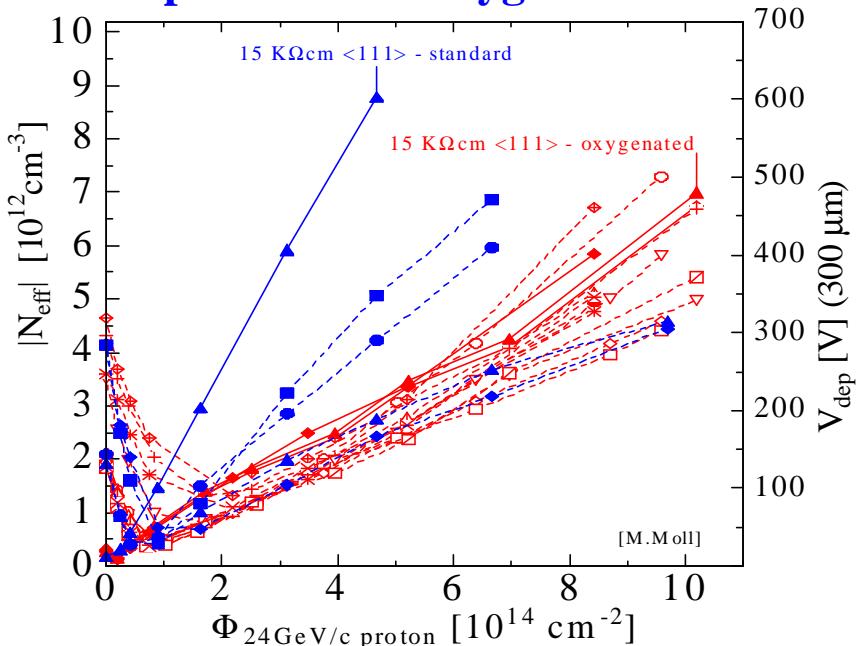
<http://cern.ch/rd48>

First tests in 1999 show clear advantage of oxygenation



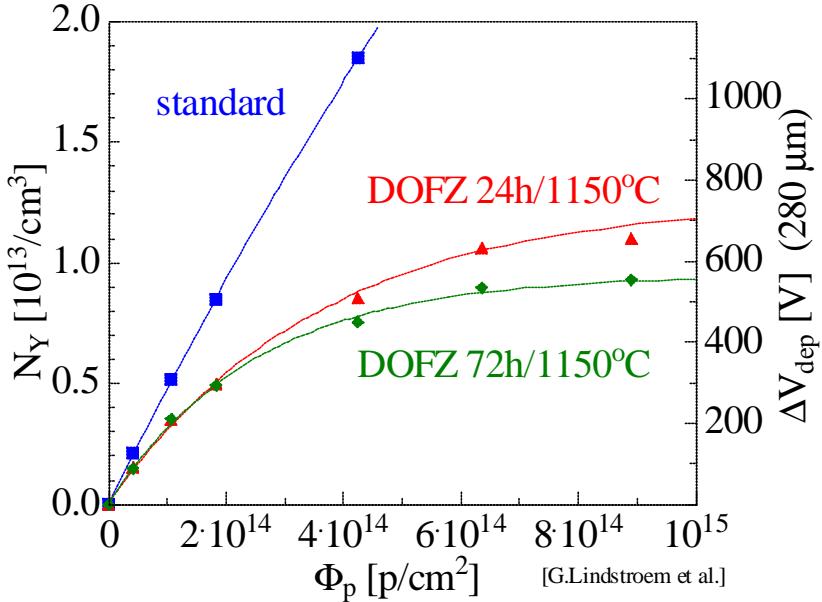
[RD48-NIMA 465(2001) 60]

Later systematic tests reveal strong variations with no clear dependence on oxygen content

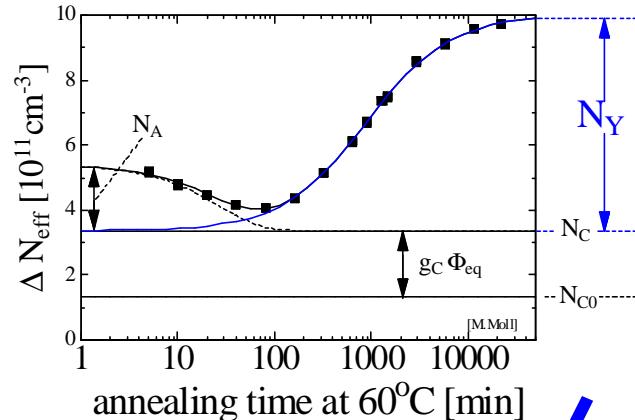


[RD50-NIMA 511 (2003) 97]

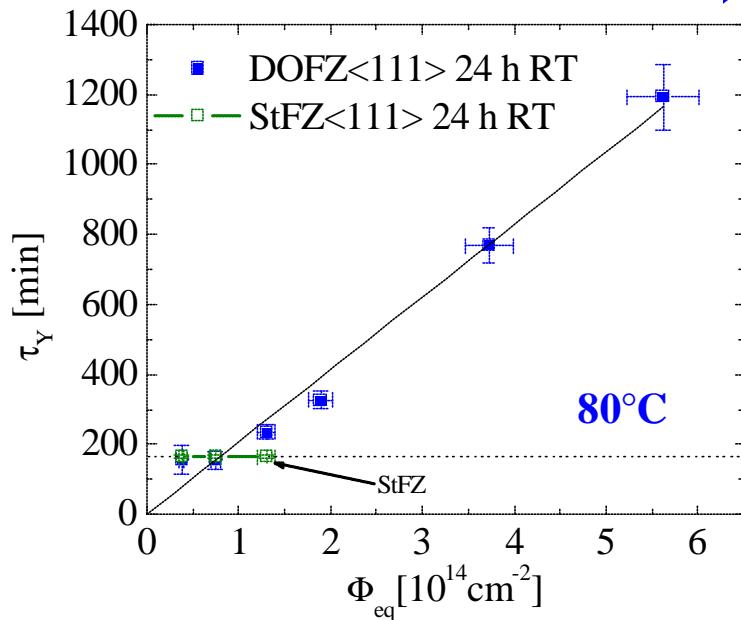
Reverse Annealing



Saturation of amplitude

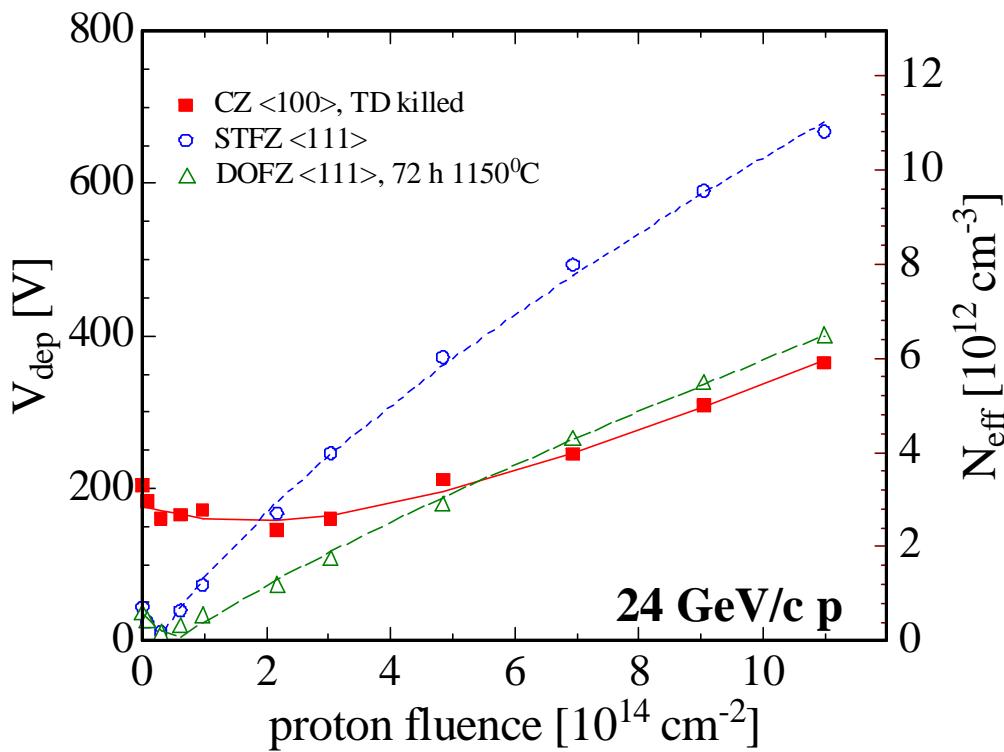


delayed reverse annealing



- **DOFZ: Saturation of reverse annealing**
(24 GeV/c p - only little effect after neutron irradiation observed !)
- **DOFZ: No big difference between 24h and 72h oxidation at 1150°C**
- **DOFZ : time constant depending on fluence**

- **Very high Oxygen content 10^{17} - 10^{18} cm $^{-3}$** (Grown in quartz (SiO_2)crucible)
- **High resistivity (>1KWcm) available only recently** (Magnetic CZ technology)

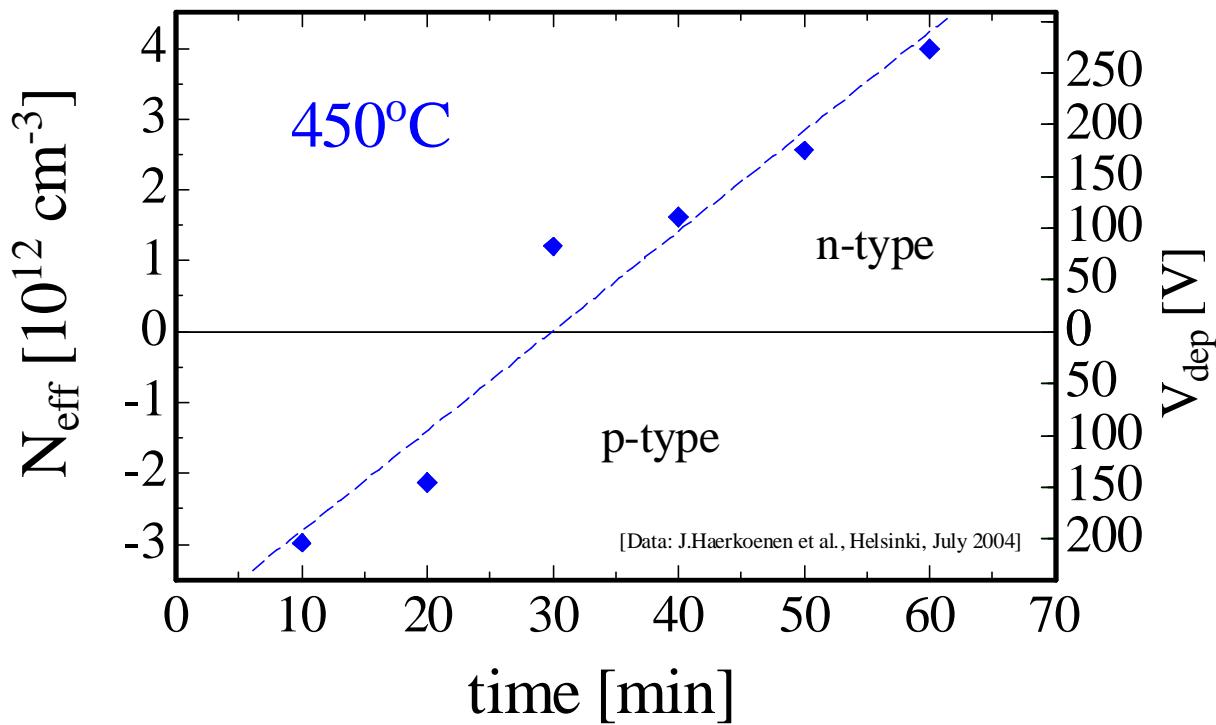


Irradiation of test-structures:

- **Only small change in V_{dep}**
 - $1 \cdot 10^{15}$ (190 MeV π)/cm 2
 - $1 \cdot 10^{15}$ (24 GeV/c p)/cm 2
 - $5 \cdot 10^{14}$ (10 MeV p)/cm 2
- **No type inversion (Sumitomo CZ)**

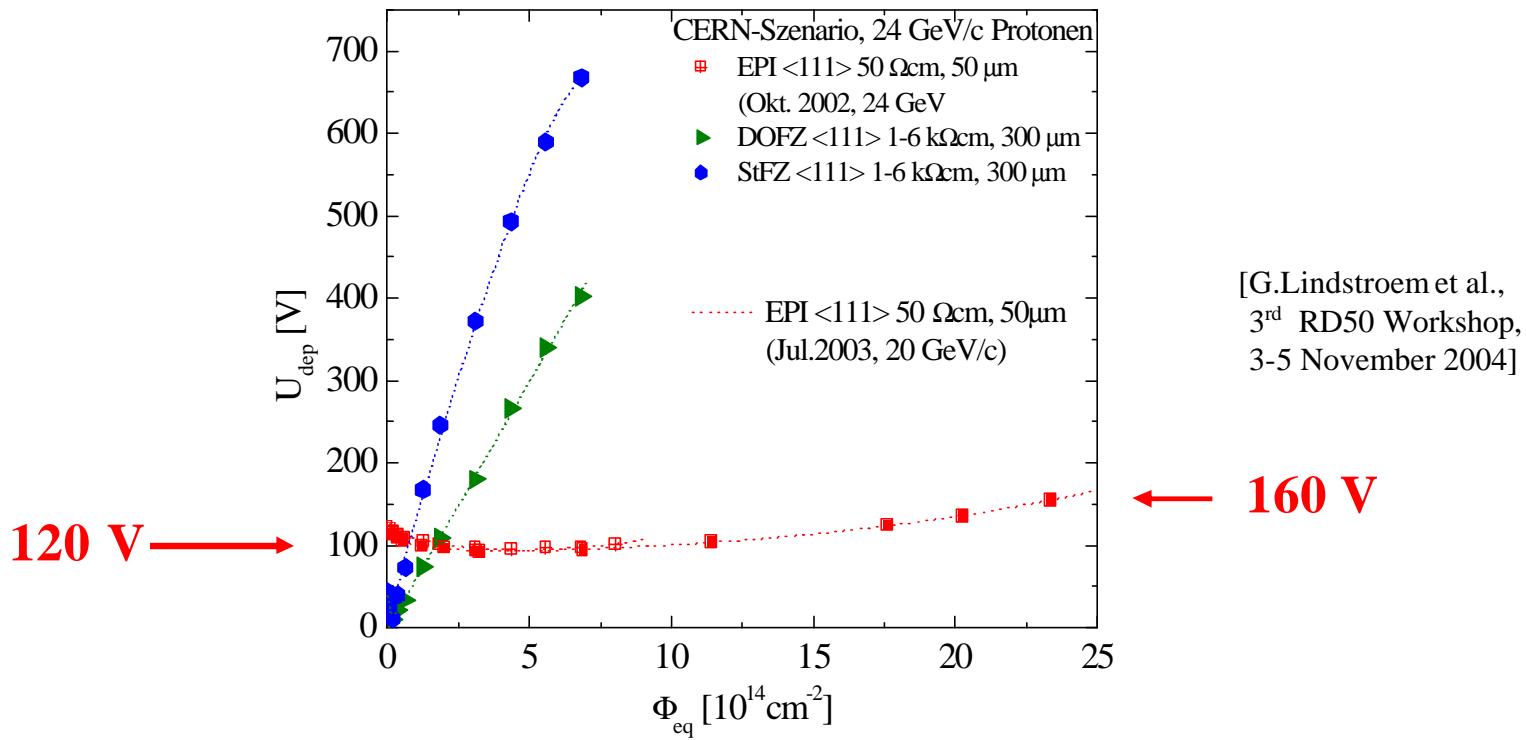
(However, type inversion observed for Okmetic MCZ after $5 \cdot 10^{14}$ (10 MeV p)/cm 2)
- **Leakage current and charge trapping as for FZ silicon**
- **Very high oxygen content:
Beware of thermal donors !**

- Thermal Donor generation due to heat treatment at 450°C
- Effective doping concentration (depletion voltage) can be tailored
 - (here: starting with p-type material and converting it to n-type)



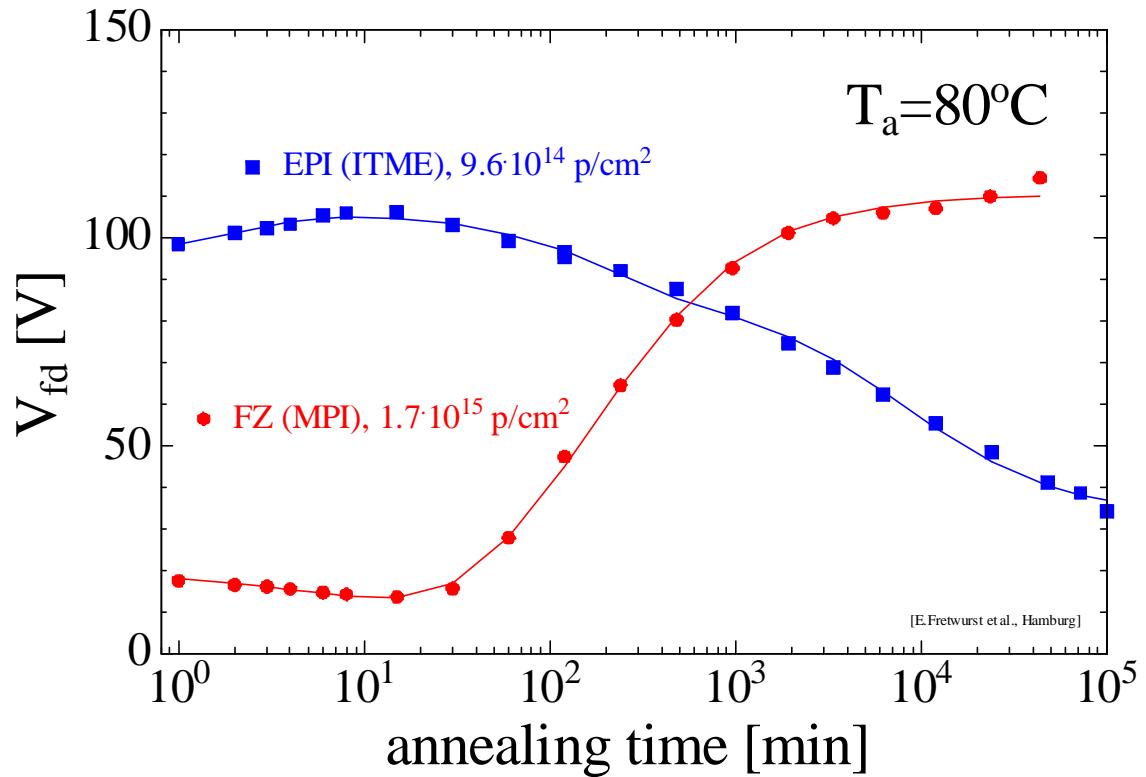
- Radiation hardness of thermal donor doped MCZ under test

- Detectors: ITME epi-silicon ($50\mu\text{m}$, $50\Omega\text{cm}$ layer on CZ-substrate) processed by CiS
- Irradiations: 24 GeV/c protons, 58 MeV Li and reactor neutrons (up to $1 \cdot 10^{16}\text{cm}^{-2}$);
no type inversion observed for proton irradiation



- Leakage current almost identical to CZ, FZ, DOFZ detectors

- 50 mm thick silicon detectors:
 - Epitaxial silicon (50Wcm on CZ substrate, ITME & CiS)
 - Thin FZ silicon (4KWcm, MPI Munich, wafer bonding technique)



[E.Fretwurst et al., 4th RD50 Workshop, 4-7 May 2004]

- **Thin FZ silicon:** Type inverted, increase of depletion voltage with time
- **Epitaxial silicon:** No type inversion, decrease of depletion voltage with time
P No need for low temperature during maintenance of SLHC detectors!

New Materials: Epitaxial SiC

“A material between Silicon and Diamond”



Property	Diamond	GaN	4H SiC	Si
E_g [eV]	5.5	3.39	3.3	1.12
$E_{breakdown}$ [V/cm]	10^7	$4 \cdot 10^6$	$2.2 \cdot 10^6$	$3 \cdot 10^5$
μ_e [cm ² /Vs]	1800	1000	800	1450
μ_h [cm ² /Vs]	1200	30	115	450
v_{sat} [cm/s]	$2.2 \cdot 10^7$	-	$2 \cdot 10^7$	$0.8 \cdot 10^7$
Z	6	31/7	14/6	14
ϵ_r	5.7	9.6	9.7	11.9
e-h energy [eV]	13	8.9	7.6-8.4	3.6
Density [g/cm ³]	3.515	6.15	3.22	2.33
Displacem. [eV]	43	315	25	13-20

- Wide bandgap (3.3eV)
- ⇒ lower leakage current than silicon

- Signal:

Diamond	36 e/mm
SiC	51 e/mm
Si	89 e/mm
- ⇒ more charge than diamond

- Higher displacement threshold than silicon
- ⇒ radiation harder than silicon (?)

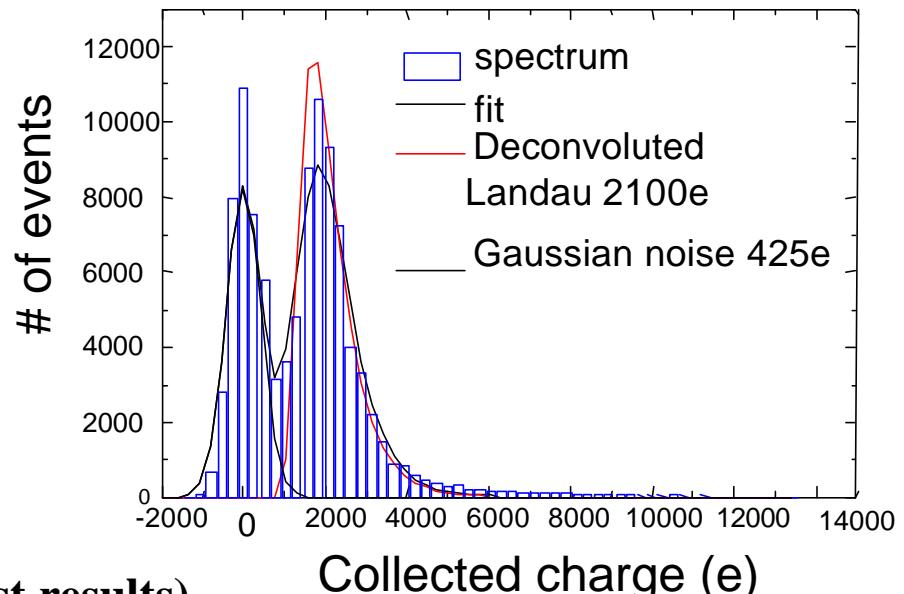
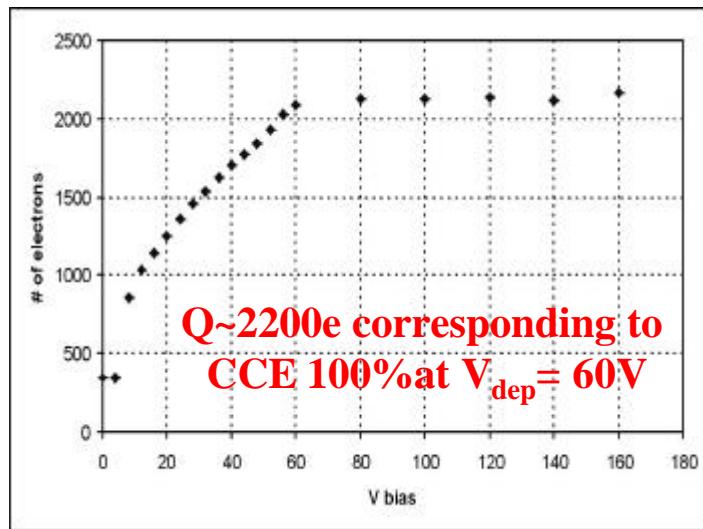
R&D on diamond detectors:

RD48 – Collaboration

<http://cern.ch/rd48/>

See talk of H.Kagan on Monday

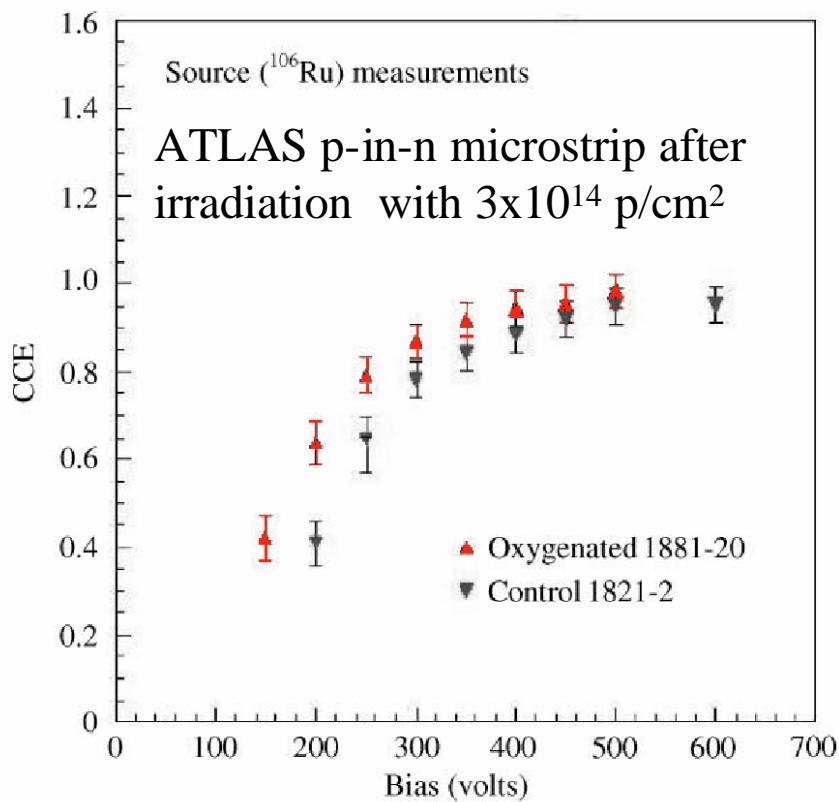
- **Semi-Insulating SiC**
 - $\rho > 10^{11} \Omega \text{cm}$ due to vanadium compensation
 - CCE 60% in as-grown, ~55% after irradiation with 10^{13}cm^{-2} 300 MeV/c π
 - Vanadium is responsible of incomplete charge collection
- **Epitaxial 4H-SiC**
 - $N_{\text{eff}} \sim 5 \cdot 10^{13} \text{cm}^{-3}$; 40 μm by IKZ Berlin on CREE substrate (Schottky contacts)



- **Irradiations: (Analysis in progress, first results)**
 - deep levels identified with DLTS (0.18-1.22eV)
 - CCE (alpha) going down to 80% after 10^{14}cm^{-2} 8MeV protons

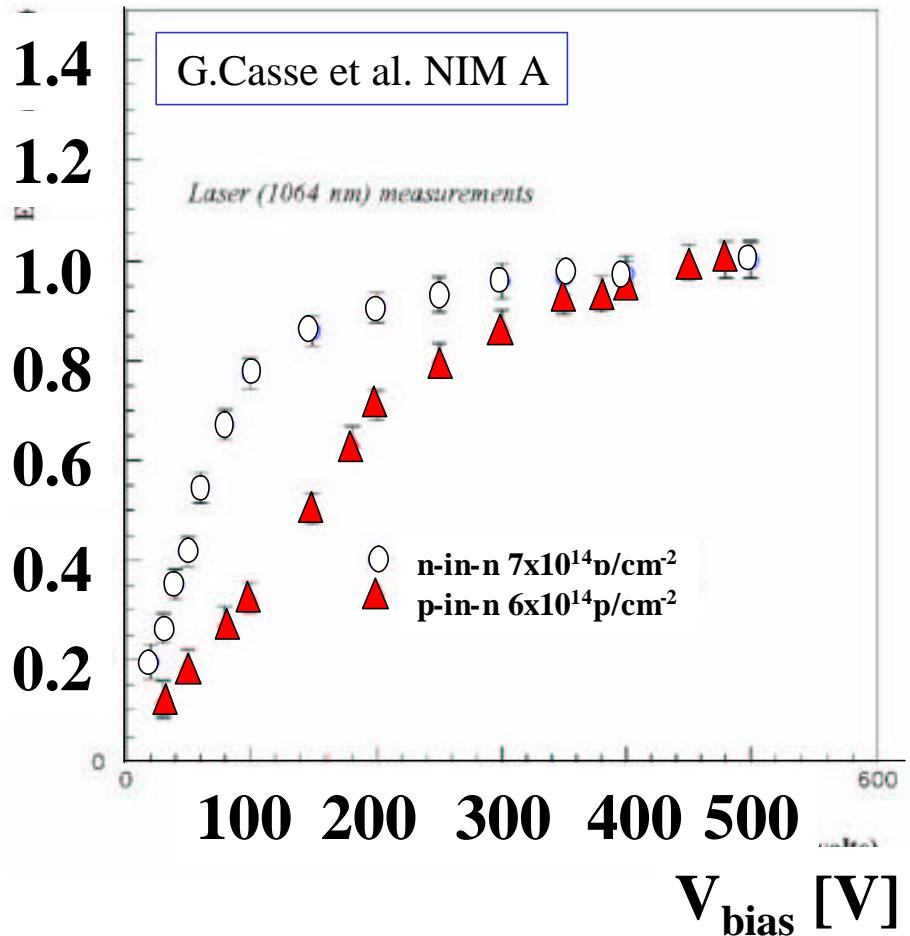
See following presentation by
James Grant (Glasgow)

- Beneficial effect of oxygen in p-irradiated DOFZ p-in-n microstrips almost disappears due to type inversion!
- Using n-in-n devices instead results in higher CCE.

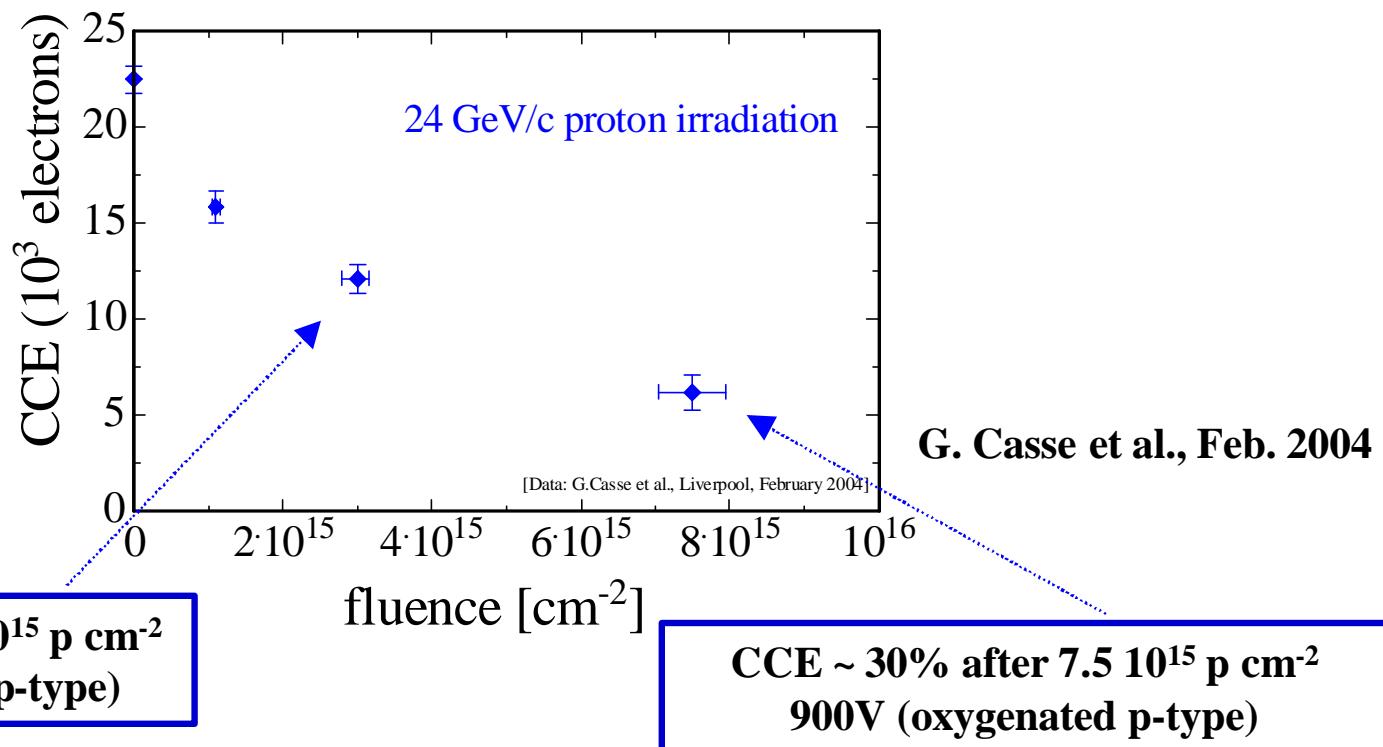


[G.Casse et al. NIM A 466 (2001) 335-344]

CCE a.u.



- Miniature n-in-p microstrip detectors (280mm).
- Detectors read-out with LHC speed (40MHz) chip (SCT128A)
- Material: standard p-type and oxygenated (DOFZ) p-type
- Irradiation:



At the highest fluence Q~6500e at $V_{\text{bias}}=900\text{V}$ corresponding to: ccd~90 μm

Motivation: After 1 MeV neutron irradiation to 10^{15} cm^{-2}
 the effective drift length for e is ~150mm and for h ~50mm

P use thin detectors (50-100mm) from the beginning

- **Benefits:**

- low operating voltage
- improved radiation tolerance: - 50µm thick, 50Ωcm Si detector ($V_{dep} = 200\text{V}$):
 - type inversion only after 10^{15} cm^{-2}

- **Drawbacks:**

- mip signal ~ 3500e-h pairs

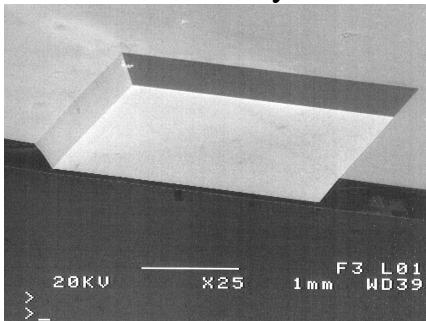
- **Technical Approaches:**

- Epitaxial Si device (shown before)
- Thinning with chemical attacks (IRST, Italy)

See following presentation by
Marco Petasecca on thin Si detectors

and wafer bonding technology (MPI Munich, Germany)

IRST: SEM of a silicon wafer thinned by TMAH



Cross section of a thinned silicon detector



Devices produced end of 2003 (up to 11mm²):

Thickness [mm]	Leakage Current [nA/cm ³]	V_{dep} [V]
300	80	12
99	30	~1
57	55	<1

- **Electrodes:**

- narrow columns along detector thickness - "3D"
- diameter: **10mm** distance: **50 - 100mm**

- **Lateral depletion:**

- lower depletion voltage needed
- thicker detectors possible
- fast signal

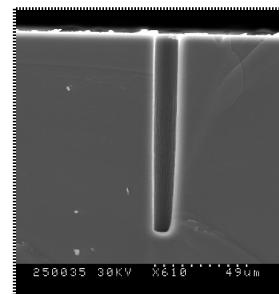
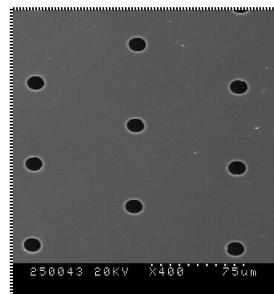
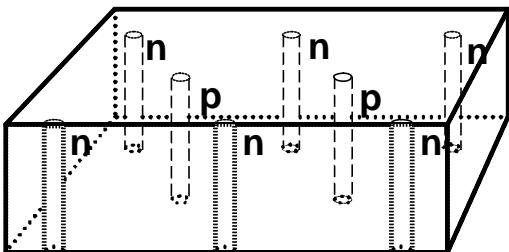
- **Hole processing :**

- Dry etching, Laser drilling, Photo Electro Chemical
- Present aspect ratio (RD50) 13:1, Target: 30:1

- **Electrode material**

- Doped Polysilicon (Si)
- Schottky (GaAs)

See overview talk by:
Sherwood Parker (Monday)



Glasgow University

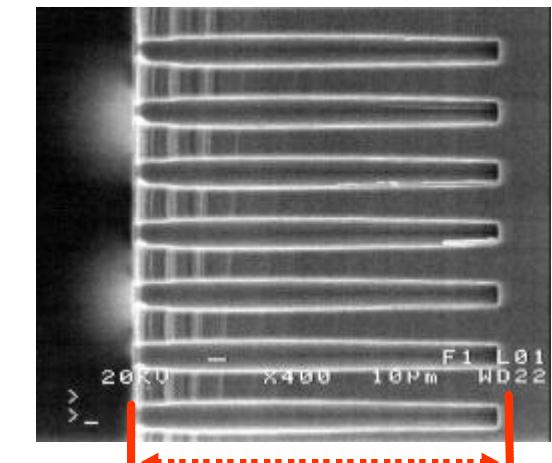
3D detector developments within RD50:

1) Glasgow University – Schottky contacts

(see Poster of Victoria Wright)

2) IRST-Trento and CNM Barcelona (since 2003)

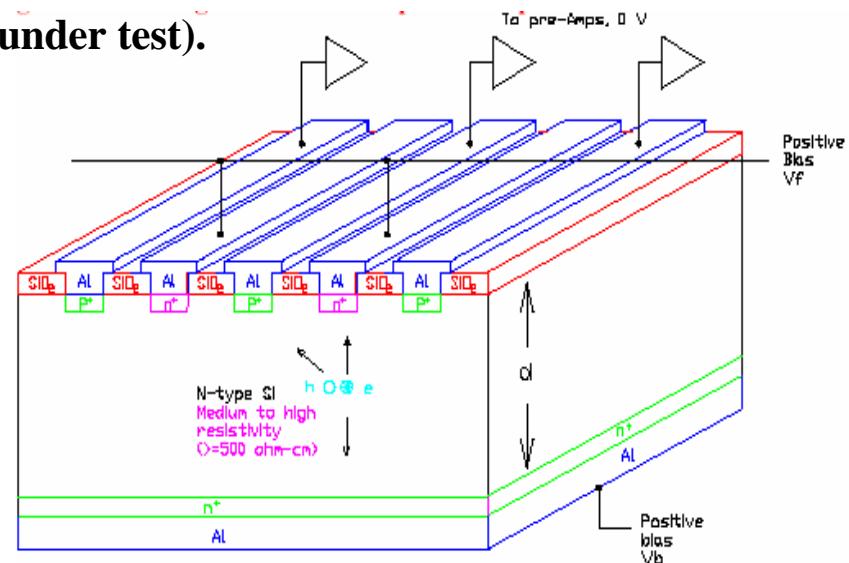
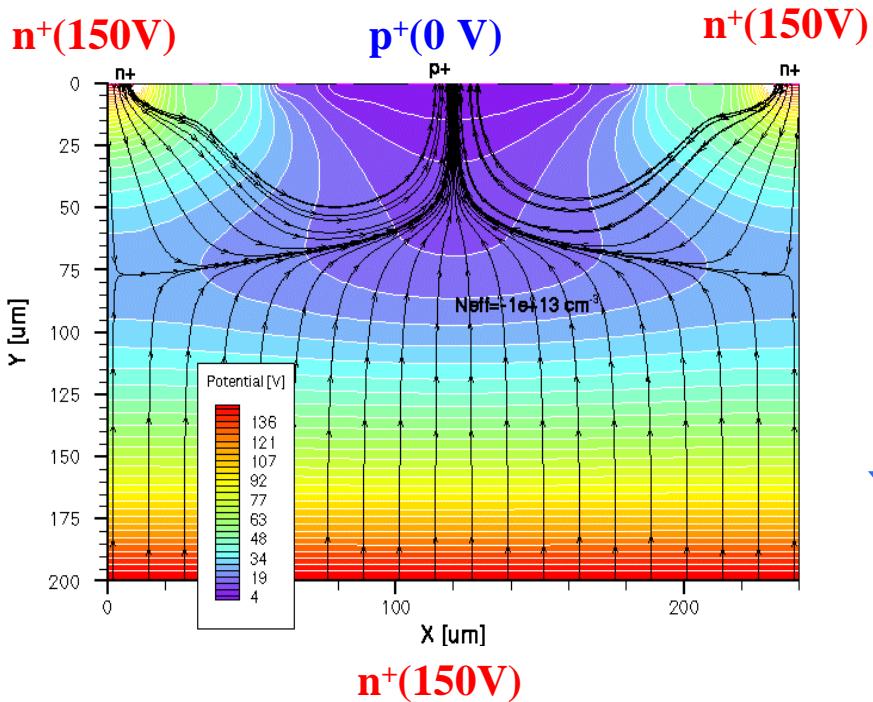
CNM: Hole etching (DRIE); IRST: all further processing
diffused contacts or doped polysilicon deposition



~200 micron

Semi 3-D devices proposed by Z. Li, BNL.

- Planar technology easier to process than 3D sensors
- Single-sided processing
- Large reduction in detector full depletion voltage after type inversion
- Processing of first prototype completed (presently under test).



Simulation of electric profile in semi 3D after irradiation to $5 \times 10^{14} \text{ n/cm}^2$.

- Radiation hard materials and new device concepts for Super LHC tracking detectors are under study by the RD50 collaboration.
- At fluences up to 10^{15}cm^{-2} (Outer layers of a SLHC detector) the change of the depletion voltage and the large area to be covered by detectors is the major problem. **CZ detectors could be a cost-effective radiation hard solution.**
- Miniature microstrip and pixel detectors made with defect engineered Si have been fabricated by RD50 or are in process. First tests with LHC like electronics are encouraging: CCE on microstrip n-in-p oxygenated detectors irradiated up to 7×10^{15} [24 GeV/c p/cm²] is > 6500 e.
- At the fluence of 10^{16}cm^{-2} (Innermost layer of a SLHC detector) the active thickness of any silicon material is significantly reduced due to trapping. The two most promising options so far are:
Thin/EPI detectors : drawback: radiation hard electronics for low signals needed
3D detectors : drawback: technology has to be optimized
- New Materials like SiC and GaN have been characterized. However, thicker samples and more radiation studies are needed to assess if these materials could be an alternative to Silicon.

Further information: <http://cern.ch/rd50/>